

# Kinetic parameters for polymerization of methyl methacrylate at 60°C

T. G. Carswell, D. J. T. Hill\*, D. I. Londero, J. H. O'Donnell, P. J. Pomery and C. L. Winzor

*Polymer and Radiation Group, Department of Chemistry, The University of Queensland, Brisbane, Queensland, Australia 4072*

*(Received 26 October 1990; revised 31 December 1990; accepted 11 January 1991)*

Fourier transform infrared spectroscopy (FTi.r.) and electron spin resonance spectroscopy (e.s.r.) have been used to follow the kinetics of the polymerization of methyl methacrylate to high conversion at 60°C. The FTi.r. absorbance at 6152 cm<sup>-1</sup> was used to monitor the time dependence of the concentration of double bonds, and the concentration of the polymer-chain propagation radicals was monitored using the e.s.r. absorption spectrum. These data were analysed to obtain instantaneous estimates of the kinetic rate parameters for propagation and termination across the range of conversion, and of the initiator efficiency at high conversion. The kinetic parameters were found to be consistent with values obtained by other methods and with the predictions of recent theories.

(Keywords: kinetic parameters; polymerization; methyl methacrylate)

## INTRODUCTION

Several models have been proposed to describe the polymerization of vinyl monomers, such as methyl methacrylate, to high conversion<sup>1-14</sup>, but these require accurate measurements of the variation in the double bond and radical concentrations with polymerization time to provide an adequate test of the description of the system. While there have been numerous studies reporting the variation in the double bond concentration, there have been relatively few studies of the corresponding variation in the radical concentration.

Here we report a study of the polymerization of methyl methacrylate to high conversion at 60°C using 2,2'-azoisobutyronitrile (AIBN) as initiator, in which we have monitored the double bond concentration by near i.r. spectroscopy, and the radical concentration by electron spin resonance (e.s.r.) spectroscopy. These data have been used to obtain instantaneous values for the propagation and termination rate constants,  $k_p$  and  $k_t$ , respectively, and for the initiator efficiency factor,  $f$ , at high conversion.

## EXPERIMENTAL

Methyl methacrylate monomer was distilled under an atmosphere of nitrogen and AIBN was recrystallized from methanol immediately before use. Oxygen was removed by bubbling the solutions with high purity nitrogen gas and then sealing. All preparation steps were carried out under minimal light conditions to avoid photoinitiation. Polymerizations were carried out in 3 mm i.d. pyrex or quartz tubes at 60°C using an initiator concentration of 0.1 mol dm<sup>-3</sup>.

Double bond concentrations were obtained, as described elsewhere<sup>15</sup>, by monitoring the first overtone

band of the C=CH vibration at 6152 cm<sup>-1</sup>, using a Mattson Sirius 100 FTi.r. spectrometer. The spectra were obtained at a resolution of 4 cm<sup>-1</sup> with a signal to noise ratio of 300, by averaging 32 interferograms in 10 s. The double bond concentration was calculated from the absorbance of the peak by application of the Beer-Lambert law.

Radical concentrations were obtained on a Bruker ER200D spectrometer using techniques described previously<sup>16</sup>. In the pregel region, polymerizations were quenched rapidly to 77 K and e.s.r. spectra were acquired subsequently at 140 K by accumulating multiple scans to enhance the signal to noise. Beyond the gel region, satisfactory spectra could be obtained from single scans when the polymerization was conducted in the cavity of the spectrometer. In the post-gel region the results obtained from single scans in the cavity and those obtained by the quenching technique were in excellent agreement.

## RESULTS AND DISCUSSION

The important kinetic steps in the bulk polymerization of vinyl monomers such as methyl methacrylate are:

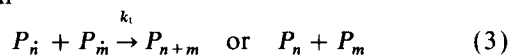
Initiation



Propagation



Termination



It follows from these chemical equations that the instantaneous rate of polymerization and the net rate of formation of radicals are given by equations (4) and (5) respectively, where  $f$  is the initiator efficiency factor, and  $[P \cdot]$  and  $[M]$  are the concentrations of polymer radicals

\* To whom correspondence should be addressed

and monomer respectively.

$$-\frac{d[M]}{dt} = k_p[P\cdot][M] \quad (4)$$

$$\frac{d[P\cdot]}{dt} = 2k_d f [I] - 2k_t [P\cdot]^2 \quad (5)$$

Thus the time dependence of the polymer radical and monomer concentrations can yield values of  $k_p$  and  $k_t$ , provided the values of  $k_d$  and  $f$  are known. The value of  $k_d$ , the rate constant for decomposition of the initiator, remains constant during polymerization, but the value of the initiator efficiency factor will begin to decrease beyond the gel point, because the probability for cage recombination of initiating radicals increases as the system changes from a liquid to a glass.

#### Monomer concentration

Fourier transform near i.r. spectroscopy offers several advantages over mid-range i.r. spectroscopy for determination of double bond concentration in vinyl polymerizations. For example, Pyrex glass cells can be used because Pyrex does not absorb in the near i.r. region of the spectrum, and path lengths of several millimetres can be employed to improve sensitivity. Scattered light effects are negligible in this region, and the accumulation function available on modern FTi.r. spectrometers provides spectra with high signal to noise ratio. A typical spectrum of methyl methacrylate is shown in Figure 1, with the C=CH band at  $6152\text{ cm}^{-1}$ . The double bond concentration can be determined from the absorbance of this peak.

Results obtained by the FTi.r. technique have been shown to agree closely with those determined by other methods<sup>17</sup>. The variation in the double bond conversion with polymerization time is shown in Figure 2. The curve shows that the rate of polymerization remains approximately constant up to a conversion of about 25% (the gel point) where the polymerizing mixture becomes very viscous. Beyond the gel point the polymerization rate increases rapidly and passes through a maximum at approximately 65% conversion. The rate of polymerization decreases rapidly beyond the glass point (approximately 80% conversion) to a small and almost constant value of approximately  $10^{-5}\text{ mol dm}^{-3}\text{ s}^{-1}$ .

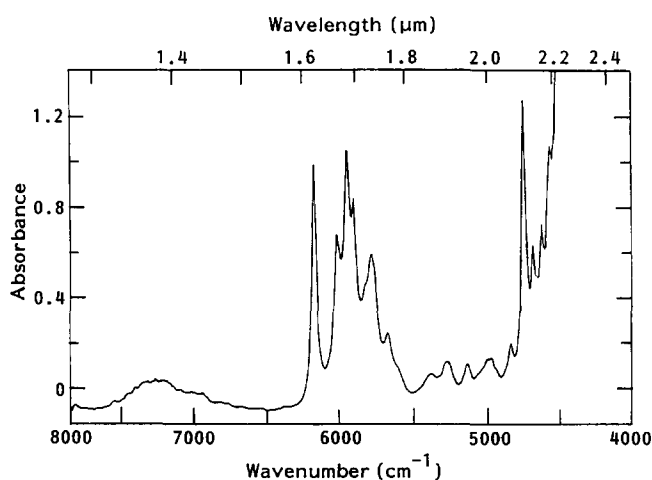


Figure 1 FTi.r. spectrum of methyl methacrylate showing the C=C absorptions at  $6152\text{ cm}^{-1}$  and  $4750\text{ cm}^{-1}$

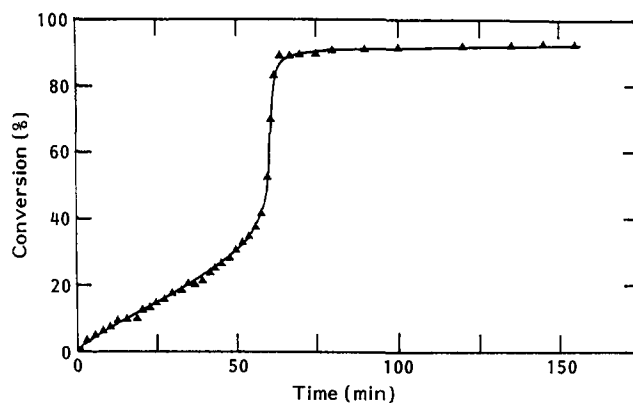


Figure 2 Plot of conversion versus time for polymerization of methyl methacrylate at  $60^\circ\text{C}$  using an AIBN concentration of 0.1 molar

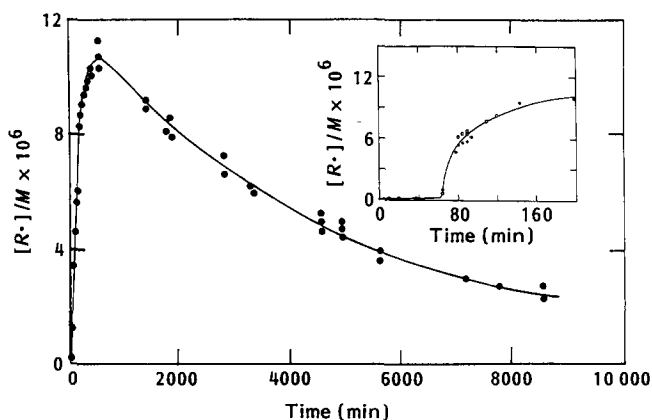


Figure 3 Plot of the radical concentration versus time for polymerization of methyl methacrylate at  $60^\circ\text{C}$  using an AIBN concentration of 0.1 molar

#### Radical concentration

The radical concentration can be determined using the e.s.r. techniques described previously<sup>16</sup>. At conversions below 25% the steady state radical concentration was too low to be obtained from a single scan, so the polymerization was rapidly quenched and the e.s.r. spectrum acquired at 140 K using accumulation of repeated scans to improve signal to noise. We have previously shown<sup>16</sup> that rapid quenching prevents any significant loss of radicals by termination during cooling. Beyond the gel point, relatively noise-free e.s.r. spectra could be obtained from single scans of a polymerizing mixture held at constant temperature in the variable temperature cavity of the e.s.r. spectrometer.

Figure 3 shows the variation in the radical concentration with time. At conversions below 25% the radical concentration remains approximately constant, with a value of  $1.5 \times 10^{-7}\text{ mol dm}^{-3}$ . Beyond the gel point the radical concentration increases rapidly until the glass point is reached, where the rate of increase in the radical concentration begins to slow. Eventually the radical concentration reached a maximum value. This occurs when the rate of initiation is equal to the rate of termination. At longer polymerization times, the radical concentration steadily falls as shown in Figure 3.

#### Kinetic parameters

Instantaneous values of the apparent rate constants  $k_p$  and  $k_t$  can be calculated from equations (4) and (5)

using the experimental information displayed in Figures 2 and 3. In the pregel region the values of  $k_p$  and  $k_t$  were found to remain approximately constant at  $510 \pm 100 \text{ dm}^3 \text{ mol}^{-1} \text{ s}^{-1}$  and  $(2.1 \pm 0.2) \times 10^7 \text{ dm}^3 \text{ mol}^{-1} \text{ s}^{-1}$  respectively. These values agree with those reported in the literature<sup>18</sup> for polymerizations to very low conversion and obtained by rotating sector methods at this temperature:  $k_p = 686 \text{ dm}^3 \text{ mol}^{-1} \text{ s}^{-1}$  and  $k_t = 3.4 \times 10^7 \text{ mol dm}^{-3} \text{ s}^{-1}$ .

Beyond the glass point the calculated value of  $k_p$  begins to decrease as shown in Figure 4. The values of  $\ln(k_p/k_{p0})$  agree closely with those predicted by Ballard *et al.*<sup>19</sup> on the basis of the theoretical model proposed by Soh and Sundberg<sup>9-12</sup>, using the assumption that the diffusion coefficient for the monomer is  $10^{-6} \text{ cm}^2 \text{ s}^{-1}$ .

The value of  $k_t$  begins to fall rapidly when the gel point is reached because the viscosity of the polymerizing mixture increases. The value of  $f$  would also decrease because the probability for recombination of the initiating radicals within the cage provided by the viscous matrix will also increase. However, once the glass point is reached, the nature of the matrix does not change very rapidly with polymerization time, because the rate of polymerization in this region is small. Thus, both  $f$  and  $k_t$  will be slowly changing functions of time in this region, and hence can be considered to be constant over short time intervals.

If equation (5) is rearranged as in equation (6) below, it follows that a plot of the left-hand side of the equation versus  $[P\cdot]^2/[I]$  should be linear over short time intervals. This relationship is demonstrated in Figure 5 for the range of polymerization times between 80 min (the glass point) and 8000 min. The instantaneous initiator concentration  $[I]$  can be calculated<sup>20</sup> at each polymerization time from the concentration of the initiator originally and the known rate constant,  $k_d$ .

$$\frac{d[P\cdot]}{dt} / [I] = 2k_d f - 2k_t [P\cdot]^2 / [I] \quad (6)$$

The plot in Figure 5 is distinctly curved, indicating that either or both  $f$  and  $k_t$  vary significantly with time

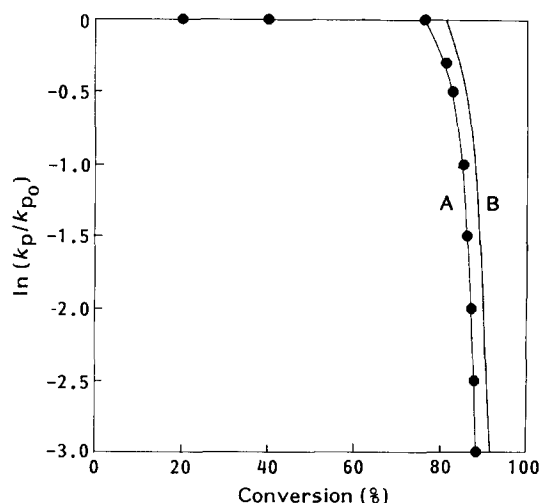


Figure 4 Plot of  $\ln(k_p/k_{p0})$  versus conversion for polymerization of methyl methacrylate at  $60^\circ\text{C}$  using an AIBN concentration of 0.1 molar: theoretical models from Ballard *et al.*<sup>19</sup> for monomer diffusion coefficients of A,  $10^{-5} \text{ cm}^2 \text{ s}^{-1}$  and B,  $10^{-6} \text{ cm}^2 \text{ s}^{-1}$ . ●, Experimental values

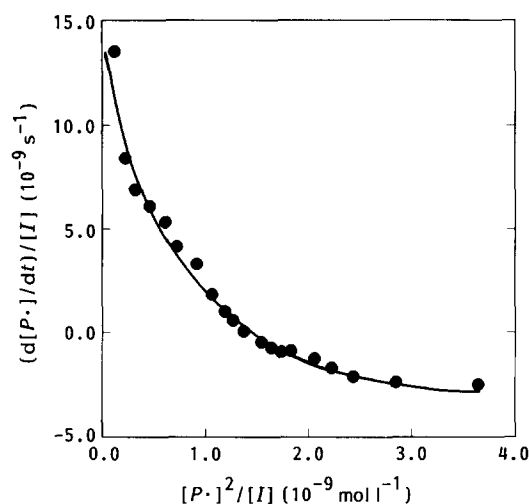


Figure 5 Analysis of experimental data using equation (6)

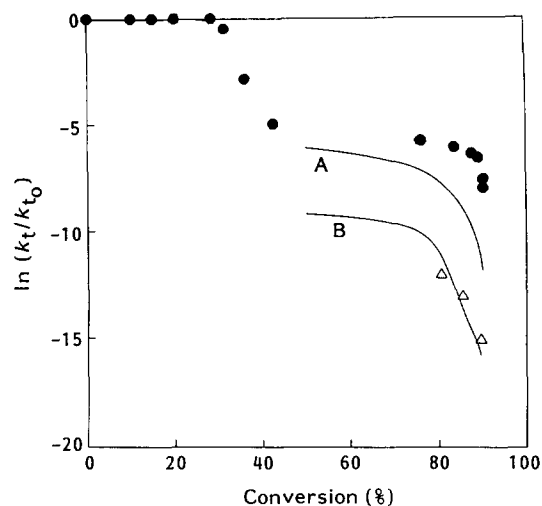


Figure 6 Plot of  $k_t/k_{t0}$  versus conversion for polymerization of methyl methacrylate at  $60^\circ\text{C}$  using an AIBN concentration of 0.1 molar: theoretical models from Russell *et al.*<sup>21</sup> for A, flexible chains and B, rigid chains. Experimental values calculated for  $f = 0.6$  (●) and  $f = 10^{-3}$  (Δ)

in this region. However, over small time intervals, it is possible to obtain estimates of the values of both  $f$  and  $k_t$  from the slope and intercept of linear sections of this plot. For example, in the region just beyond the glass point (corresponding to polymerization times between 100 and 3000 min) the values calculated for  $f$  and  $k_t$  are  $10^{-3}$  and  $3.5 \text{ dm}^3 \text{ mol}^{-1} \text{ s}^{-1}$ , respectively. The monomer conversion over this range of polymerization times (between 100 and 1000 min) is 90–95%.

This value of  $k_t$  is approximately seven orders of magnitude smaller than that found in the pregel region,  $2.1 \times 10^7 \text{ dm}^3 \text{ mol}^{-1} \text{ s}^{-1}$ . The large decrease in  $k_t$  beyond the gel point is consistent with the behaviour predicted by recent theoretical models, see Figure 6. The value obtained for  $f$  in the region beyond the glass point is also much less than that for the pregel region. This observation is also consistent with the predictions based upon theoretical models, although the value calculated here is somewhat larger than that predicted by Russell *et al.*<sup>21</sup> for polymerizations of methyl methacrylate at  $50^\circ\text{C}$ .

Buback<sup>14</sup> has recently reported a modification of the theory of Russell *et al.*<sup>21</sup> in which he expresses the residual termination rate constant,  $k_{tRD}$ , (which is equivalent to  $k_t$  in the glassy region) in terms of the propagation rate constant,  $k_p$ , and the fraction of double bonds remaining ( $1 - U$ ):

$$k_{tRD} = C_{RD}k_p(1 - U) \quad (7)$$

The proportionality constant  $C_{RD}$  was calculated by Buback to be approximately 187 at 60% conversion for polymerizations of methyl methacrylate, and fairly independent of temperature. Calculation of  $C_{RD}$  using the data reported herein at 60% conversion yields a value of 276. This value is of similar magnitude to that reported by Buback, which was based on data for emulsion polymerizations at 50°C and bulk polymerizations at 0°C. Considering the likely errors in the experimental data, the two estimates of  $C_{RD}$  can be considered to be in reasonable agreement.

#### SUMMARY

The kinetic parameters for the polymerization of methyl methacrylate have been determined at 60°C using FTi.r. spectroscopy to measure the conversion and e.s.r. spectroscopy to measure the propagating radical concentration over the range from zero to approximately 95% conversion. The initiator efficiency has been shown to be much smaller in the post-glass region than it is in the pregel region of the polymerization.

#### ACKNOWLEDGEMENT

The authors thank the Australian Research Committee for supporting this research.

#### REFERENCES

- 1 Cardenas, J. N. and O'Driscoll, K. F. *J. Polym. Sci., Polym. Chem. Edn* 1976, **14**, 883
- 2 Cardenas, J. N. and O'Driscoll, K. F. *J. Polym. Sci., Polym. Chem. Edn* 1977, **15**, 1883
- 3 Dionisio, J. M. and O'Driscoll, K. F. *J. Polym. Sci., Polym. Chem. Edn* 1980, **18**, 241
- 4 O'Driscoll, K. F. *Pure Appl. Chem.* 1981, **53**, 617
- 5 Martin, F. L. and Hamielec, A. E. *Am. Chem. Soc. Symp. Ser.* 1979, **104**, 43
- 6 Tulig, T. J. and Tirrell, M. *Macromolecules* 1981, **14**, 1501
- 7 Tulig, T. J. and Tirrell, M. *Macromolecules* 1982, **15**, 459
- 8 Panke, D., Stickler, M. and Wunderlich, W. *Makromol. Chem.* 1983, **184**, 175
- 9 Soh, S. K. and Sundberg, D. C. *J. Polym. Sci., Polym. Chem. Edn.* 1982, **20**, 1299
- 10 Soh, S. K. and Sundberg, D. C. *J. Polym. Sci., Polym. Chem. Edn.* 1982, **20**, 1315
- 11 Soh, S. K. and Sundberg, D. C. *J. Polym. Sci., Polym. Chem. Edn* 1982, **20**, 1331
- 12 Soh, S. K. and Sundberg, D. C. *J. Polym. Sci., Polym. Chem. Edn.* 1982, **20**, 1345
- 13 Chiu, W. Y., Carratt, G. M. and Soong, D. S. *Macromolecules* 1983, **18**, 348
- 14 Buback, M. *Makromol. Chem.* 1990, **191**, 1575
- 15 Carswell, T. G., Hill, D. J. T., O'Donnell, J. H., Pomery, P. J. and Winzor, C. L. Preprints of papers presented at IUPAC International Symposium, Molecular Design of Functional Polymers, Seoul, Korea, 1986, p. 286
- 16 Carswell, T. G., Hill, D. J. T., Hunter, D. S., O'Donnell, J. H., Pomery, P. J. and Winzor, C. L. *Eur. Polym. J.* 1990, **26**, 541
- 17 Winzor, C. L. *PhD Thesis* University of Queensland, Brisbane, 1989
- 18 Mahabadi, H. K. and O'Driscoll, K. F. *J. Makromol. Sci., Chem.* 1977, **A11**, 967
- 19 Ballard, M. J., Gilbert, R. G., Napper, D. H., Pomery, P. J., O'Sullivan, P. W. and O'Donnell, J. H. *Macromolecules* 1986, **19**, 1303
- 20 Moroni, A. F. *Makromol. Chem.* 1967, **105**, 43
- 21 Russell, G. T., Napper, D. H. and Gilbert, R. G. *Macromolecules* 1988, **21**, 2141