polymer communications

Re-examination of a proposed method to obtain Avrami parameters directly from non-isothermal crystallization data

Jinan Cao

CSIRO Division of Coal & Energy Technology, PO Box 136, North Ryde, NSW 2113, Australia (Received 15 May 1991; revised 29 January 1992)

This communication is a theoretical re-examination of the method to obtain Avrami parameters directly from non-isothermal crystallization data proposed by Dutta.

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Introduction

The crystallization kinetics of a polymer are important in the elucidation of the crystallization mechanism and have been studied extensively. However, in the recent article by Dutta1 entitled 'Method to obtain Avrami parameters directly from non-isothermal crystallization data', some important aspects were disregarded. These aspects will be addressed below.

Methodology

As noted by Dutta, the Avrami equation and the crystallization rate equation [Dutta's equations (1) and (3)] are applicable only for an isothermal crystallization. The temperature dependence of the crystallization rate constant k is, in this case, a comparison of crystallization rates of different isothermal crystallization processes. The isokinetic assumption which suggests that a nonisothermal crystallization can be expressed as the superposition of a sequence of small isothermal crystallization steps, does not mean that a non-isothermal crystallization may be dealt with as an isothermal process even in the case of a linear temperature change^{2,3}. Dutta's non-isothermal equation (7) was derived by combining the effect of temperature on the crystallization rate constant k and the isothermal crystallization rate $\dot{\alpha}$ in equation (3), then differentiating $\dot{\alpha}$ with respect to time t. This treatment ignored the physical meaning of the Avrami equation, indicating a methodological error.

This error becomes more obvious if it is noted mathematically that when equation (7) was derived from equation (3), $dk/dt = -k\beta E'/T^2$ was employed; on the other hand, dk/dt = 0 was adopted in the derivation of equation (3) from equation (1) since an isothermal crystallization has been assumed.

Pre-exponential factor

The incorrectness of the method proposed by Dutta may be further demonstrated if one realizes that a different pre-exponential factor A will be obtained following the method. When the condition $d^2\alpha/dt^2=0$ is satisfied, the crystallization rate reaches its maximum. Substituting Dutta's equation (7) into Dutta's equation

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(3) leads to:

$$A = \frac{\dot{\alpha}_{p}(T_{p} - T_{0})}{(1 - \alpha_{p})\beta} \frac{1}{\left(\frac{T_{p} - T_{0}}{\beta}\right)^{n}} \exp\left(-\frac{E'}{T_{p}}\right) \frac{1}{n}$$
(1)

where all the symbols have the same meanings as those used by Dutta.

If the method is correct, a consistent A would be anticipated. Table 1 shows the values of A calculated according to the above equation as well as the peak temperatures T_p calculated from the numerical readings of Dutta's Figure 2. It can be seen that the variation in A is too large to be acceptable. A contradictory pre-exponential factor will necessarily lead to an erroneous rate constant.

Non-isothermal crystallization and crystallization rate constant

It is worth mentioning that instead of the Avrami equation, the following generalized equation should be used for a non-isothermal crystallization under the isokinetic assumption:

$$\alpha = 1 - \exp\left(-\left\{\int_{0}^{t} k[T(\tau)]^{1/n} d\tau\right\}^{n}\right)$$
 (2)

where τ denotes the time variable of the integrand, and other symbols have the same meanings as those used by Dutta.

The effect of temperature on the crystallization rate constant k of a polymer is complicated. Even when discussion is limited to the case of poly(ethylene

Table 1 Values of A calculated from equation (1)

$\beta(\mathbf{K} min^{-1})$	$\frac{T_0-T_p}{T_p^2}(\mathbf{K}^{-1})$	$\frac{\dot{\alpha}_{p}(T_{p}-T_{0})}{(1-\alpha_{p})\beta}$	$T_{p}(\mathbf{K})$	A (min ⁻ ")
-2	0.000419	18.167	477.6	2.6×10^{-40}
-3	0.000436	18.905	474.8	4.2×10^{-40}
-5	0.000459	19.762	471.2	7.2×10^{-40}

 $T_0 = 573.1 \text{ K}, E' = 40022.4 \text{ K}^{-1}, n = 2.42$

terephthalate) (PET), the Arrhenius relationship fails to interpret why a slow crystallization is observed when a PET melt is cooled to close to its glass transition temperature. It is well known that the crystallization rate dependence on temperature can be represented by a bell-shaped curve. The crystallization half-time of Gaussian type is given by^{3,4}:

$$\frac{1}{t_{1/2}} = \frac{1}{t_{1/2}^*} \exp\left[-\frac{4\ln(2(T-T^*)^2)}{D^2}\right]$$
 (3)

where $t_{1/2}^*$ is the minimum half-time corresponding to the maximum crystallization rate constant at the temperature

 T^* , and D is the width at half-height of the Gaussian curve. This relationship is certainly incompatible with the Arrhenius equation.

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